



## Surface plasmon assisted control of hot-electron relaxation time

SARVENAZ MEMARZADEH,<sup>1,2,†</sup> JONGBUM KIM,<sup>2,3,4,†</sup> YIGIT AYTAÇ,<sup>5</sup> THOMAS E. MURPHY,<sup>1,2</sup> AND JEREMY N. MUNDAY<sup>1,2,3,\*</sup>

<sup>1</sup>Department of Electrical and Computer Engineering, University of Maryland, College Park, Maryland 20742, USA

<sup>2</sup>Institute for Research in Electronics and Applied Physics, University of Maryland, College Park, Maryland 20742, USA

<sup>3</sup>Department of Electrical and Computer Engineering, University of California, Davis, California 95616, USA

<sup>4</sup>Nanophotonic Research Center, Korea Institute of Science and Technology (KIST), Seoul 02792, South Korea

<sup>5</sup>Science Systems and Applications, Inc., Lanham, Maryland 20706, USA

\*Corresponding author: jnmunday@ucdavis.edu

Received 16 December 2019; revised 20 April 2020; accepted 25 April 2020 (Doc. ID 385959); published 27 May 2020

Surface plasmon mediated hot-carrier generation is utilized widely for the manipulation of electron–photon interactions in many types of optoelectronic devices including solar cells, photodiodes, and optical modulators. A diversity of plasmonic systems such as nanoparticles, resonators, and waveguides has been introduced to enhance hot-carrier generation; however, the impact of propagating surface plasmons on hot-carrier lifetime has not been clearly demonstrated. Here, we systematically study the hot-carrier relaxation in thin film gold (Au) samples under surface plasmon coupling with the Kretschmann configuration. We observe that the locally confined electric field at the surface of the metal significantly affects the hot-carrier distribution and electron temperature, which results in a slowing of the hot electrons' relaxation time, regardless of the average value of the absorbed power in the Au thin film. This result could be extended to other plasmonic nanostructures, enabling the control of hot-carrier lifetimes throughout the optical frequency range. © 2020 Optical Society of America under the terms of the [OSA Open Access Publishing Agreement](https://doi.org/10.1364/OPTICA.385959)

<https://doi.org/10.1364/OPTICA.385959>

### 1. INTRODUCTION

Recently, the optical generation of hot carriers in metallic components has attracted interest for applications such as solar energy conversion [1–5], nonlinear optics [6–8], sensitive photodetectors [9–12], nanoscale heat sources [13], photochemical reactions in biomolecular studies [14–16], and biosensors [17,18]. For the excitation of hot carriers in metals, the incident photon energy is typically lower than the energy of the band-to-band transition, and thus the efficiency of hot-carrier generation is reduced as a result of the poor absorption of light within the metals. To overcome this limitation, surface plasmons have been broadly utilized to enhance absorption through the use of metallic nanostructures [19–23], which increases the measurement sensitivity because of the increased absorption [24]. Furthermore, the epsilon-near-zero (ENZ) mode in metallic semiconductors has also recently been used to improve the photon–electron interaction for enhancement of hot-carrier generation [25,26]. Hot carriers relax to equilibrium through plasmon dephasing via Landau damping, electron–electron (e–e) scattering, electron–phonon (e–ph) scattering, and lattice heat dissipation through phonon–phonon (ph–ph) interactions [27]. Throughout these processes, hot carriers can distribute their energy to the surrounding environment and in turn thermalize from their excited state to equilibrium. The temporal duration of hot-carrier relaxation is the key factor to determine

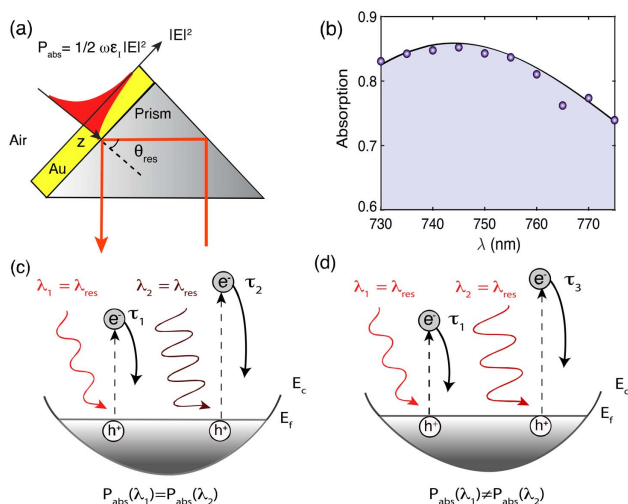
the performance of hot-carrier devices. For example, the efficiency of hot-carrier injection in energy conversion systems [5,28] and the operating speed in optical modulation systems [29,30] are both strongly linked to hot-carriers' lifetime. Depending on the geometry of metal nanostructures, the materials' band structure, and the incident photons' energy [21,31], the relaxation time can vary from a few hundred femtoseconds up to a couple of picoseconds. In the case of gold (Au) and aluminum nanostructures, relaxation times on the order of hundreds of picoseconds, due to the acoustic vibrations of the lattice, have been reported [32–34]. The effect of enhanced absorption on hot-carrier relaxation time has been studied extensively in the case of thin film and nanostructured plasmonic systems [35–37]; however, the importance of the strongly confined field inside the metal thin film induced by surface plasmon coupling on hot-carrier lifetimes is still elusive. Transient reflectivity measurements using pump–probe spectroscopy are a common method to characterize carrier dynamics under the intra-band or inter-band transitions. Typically, the measured transient signals for pump–probe spectroscopy are analyzed with the two-temperature model (TTM), which describes the spatiotemporal profile of the electron and the lattice temperature from a coupled nonlinear partial differential equation [38–40]. This model is very useful in understanding relaxation dynamics, but appropriate modification is needed for an accurate modeling

of the unique internal electric field profile in metal films due to its coupling to the propagating surface plasmon. In this work, we experimentally investigate the relationship between the hot-carrier relaxation time and the characteristics of surface plasmons on gold (Au) thin films excited under the Kretschmann configuration. For accurate theoretical modeling of the transient reflectivity data resulting from the carrier dynamics in the conduction band of Au thin film, we employ the free electron model to estimate the elevated electron temperature due to intra-band optical pumping. From the calculated electron temperature, we extract the carrier relaxation time with the modified TTM to better describe the localized electric field distribution inside the Au thin film. Under fixed absorbed power in the Au film over the spectral range of 730 nm to 775 nm (resonance wavelength at 745 nm), we observe that the hot-electron relaxation time in the Au film reaches its maximum at the resonance wavelength, which indicates that the modified intensity and profile of the internal electric field by the excitation of surface plasmons play a significant role in hot-carrier relaxation.

## 2. RESULTS AND DISCUSSION

To study the effect of surface plasmons on hot-electron relaxation dynamics, we combine the prism coupling technique under the Kretschmann configuration, schematically illustrated in Fig. 1(a), with pump–probe optical spectroscopy.

A 10 mm N-BK7 right angle prism with an AR coating (wavelength range 650–1050 nm) on the face of the hypotenuse is used for the prism coupling. An e-beam evaporator is employed for the Au deposition at a starting pressure of  $3 \times 10^{-6}$  Torr. The permittivity of the Au film is characterized by a Woollam-M2000 spectroscopic ellipsometer and is applied for the optical simulation.



**Fig. 1.** (a) Schematic of light coupling to propagating surface plasmons using the Kretschmann configuration. (b) Absorption measurement (circles) and simulation (solid line) after surface plasmon coupling. (c) Schematic diagram showing hot-electron excitation under resonance and off-resonance wavelengths while keeping the absorbed power fixed (120 mW) for both illuminations. (d) Schematic diagram showing a second case where the hot-electron excitation occurs under the same resonance wavelength (745 nm) but with different absorbed powers.  $\tau_1$ ,  $\tau_2$ , and  $\tau_3$  are the corresponding electron–phonon relaxation times for these different cases.

We use a precise motorized rotational mount with 25 arcsec angular resolution for coupling to the propagating surface plasmon. The incident beam from the glass interface is focused on the Au side of the prism using an off-axis parabolic mirror. Both reflection and transmission of the incoming beam are recorded while rotating the prism automatically. Transmission of the sample is measured to be less than 1% and therefore is negligible for determination of the absorption ( $A = 1 - R$ ). To incorporate possible scattering effects from every interface of the prism, we optimize our absorption measurement using a bare prism first, without any Au coating, to measure the baseline of the reflection signal. The bare prism is then replaced by the Au-coated prism on the rotational stage for the surface plasmon coupling. The reflection signal is recorded over the incident angle for different pump wavelengths. The final signal is the ratio between the reflectivity measured using coated and uncoated prisms.

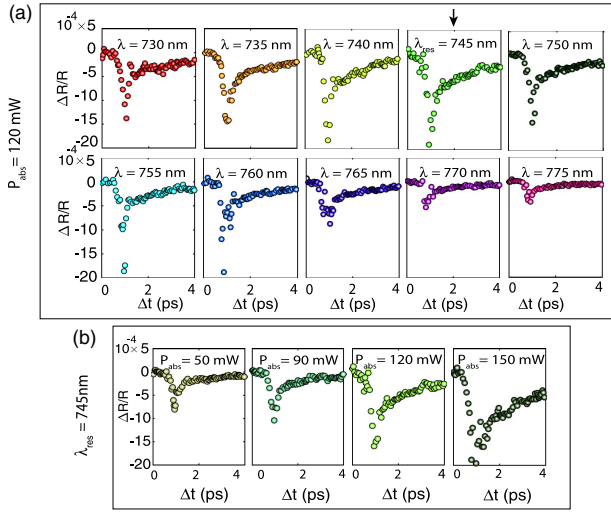
The thickness of the Au film and the incident angle of light are set to 44 nm and  $44^\circ$ , respectively. Under these conditions, surface plasmon excitation occurs at 745 nm (1.66 eV), where the photon energy is lower than the d-band transition of Au, at 2.4 eV [41]. Once the surface plasmon is excited in the Au film, the electric field is strongly confined at the interface between the Au film and air. Figure 1(b) shows absorption as a function of wavelength ranging from 730 nm to 775 nm, with resonance wavelength at 745 nm (see Fig. S1 of Supplement 1 for the broad range of absorption spectrum).

For the time-resolved differential reflectivity measurements, we employ a degenerate pump–probe technique. Transverse-magnetic (TM) polarized pulses are produced from a femtosecond Ti-sapphire laser with 80 MHz repetition rate. Using a beam splitter, the incoming pulses are then separated into pump and probe paths. Both beams are directed to coincide on the Au surface after reflecting off the off-axis parabolic mirror to a spot size of approximately  $40 \mu\text{m}$ . To optimize the signal, the overlap between the two beams is monitored using an AmScope MU1000 digital microscope camera. After spatially separating the two beams, the probe beam is then directed to the Si photodetector for differential reflectivity measurements. The time delay between the pump and probe pulses is produced by passing the pump beam through the mechanical delay stage.

To rule out the effect of absorbed light power in the control of the hot-carrier relaxation temporal dynamics, we designed two different experimental conditions: (1) sweeping the wavelength ( $\lambda = 730 \sim 775$  nm) with fixed absorbed power ( $P_{\text{abs}} = 120$  mW), and (2) varying the absorbed laser power ( $P_{\text{abs}} = 50 \sim 150$  mW) with a fixed wavelength ( $\lambda = 745$  nm). Figures 1(c) and 1(d) schematically illustrate hot-electron excitation under these two conditions.

Transient reflectivity ( $\Delta R/R_0$ ) measurements as a function of time delay ( $\Delta t$ ) between the pump and probe for both conditions are shown in Figure 2. When the wavelength is varied, we adjust the incident pump intensity according to the absorption spectra [Fig. 1(b)] to ensure that the absorbed power remains the same over the entire incident wavelength range. We observe that the transient reflectivity ( $\Delta R/R_0$ ) reaches the maximum at resonance, and signal modulation is gradually reduced as the wavelengths tend away from resonance. For the case of fixed wavelength illumination, the input power is varied (59 mW, 105 mW, 141 mW, and 176 mW) at the resonance wavelength.

Transient reflectivity ( $\Delta R/R_0$ ) can be converted to the electron temperature under the intra-band optical pumping,



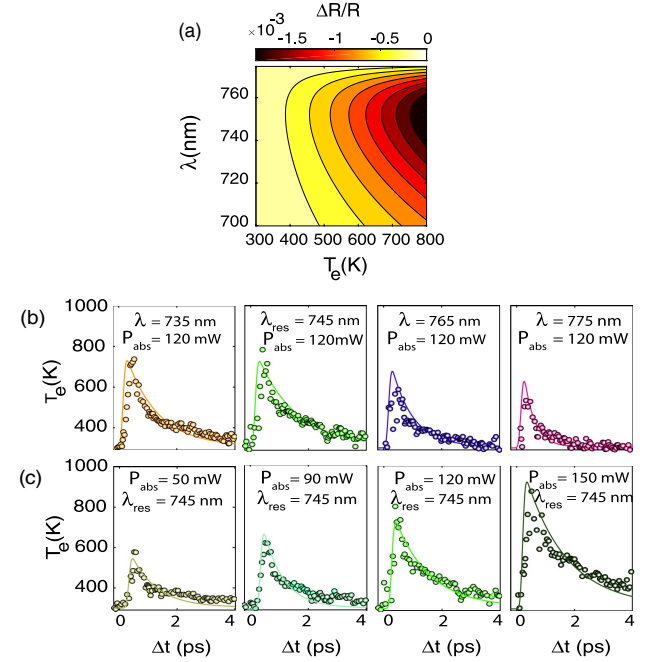
**Fig. 2.** (a) Relative reflectivity change for different incident wavelengths ranging from 730 nm to 775 nm measured at fixed absorbed power (120 mW). Resonance wavelength is distinguished by a green frame from the rest of the wavelengths. (b) Relative reflectivity signals under fixed 745 nm resonance wavelength measured with different absorbed powers (50 mW, 90 mW, 120 mW, 150 mW).

which results in a nonequilibrium hot-electron distribution that can modify the optical properties of the Au film. The Au band structure is modeled using a simplified parabolic electron density of states [42]. Considering that the carrier density is a temperature-independent quantity and the intra-band excitation does not generate extra carriers in the conduction band ( $N_{e,pump} = N_{e,nopump} = 5.049 \times 10^{22} \text{ cm}^{-3}$ ), we can calculate the chemical potential, Drude plasma frequency ( $\omega_p = \sqrt{(e^2 N_e) / (\epsilon_0 \epsilon_\infty m_e^* m_0)}$ ), and damping coefficient ( $\Gamma_p = \hbar e / (m_e^* m_0 \mu_e)$ ) as a function of the electron temperature, where  $N_e$  is the carrier concentration,  $m_0$  is the mass of electron,  $m_e^*$  is the dimensionless effective electron mass, and  $\mu_e$  is the electron mobility (see Supplement 1 for free electron model details). Subsequently, the change in reflectivity with electron temperature over different incident wavelengths can be determined from the transfer matrix method (TMM) calculation, as shown in Fig. 3(a). Typically, in order to extract the relaxation time of a nonequilibrium system, direct fitting of the TTM is applied to the transient spectroscopic measurements [43,44]. However, it is noted that the change in reflectivity is not linearly proportional to the electron temperature; hence, for clear comparison and to better estimate the hot-carrier relaxation dynamics, the direct fitting of the TTM is performed on the electron temperature extracted from the transient reflectivity measurements.

Figures 3(b) and 3(c) show the converted electron temperature as a function of time delay for both fixed absorbed power with varied wavelengths, and fixed resonance wavelength with varied absorbed powers. The converted electron temperatures can be modeled using the TTM as a function of depth from surface ( $z$ ) and time ( $t$ ) as follows:

$$C_e(T_e) \frac{\partial T_e}{\partial t} = K_e \nabla^2 T_e - G(T_e - T_l) + S(z, t), \quad (1)$$

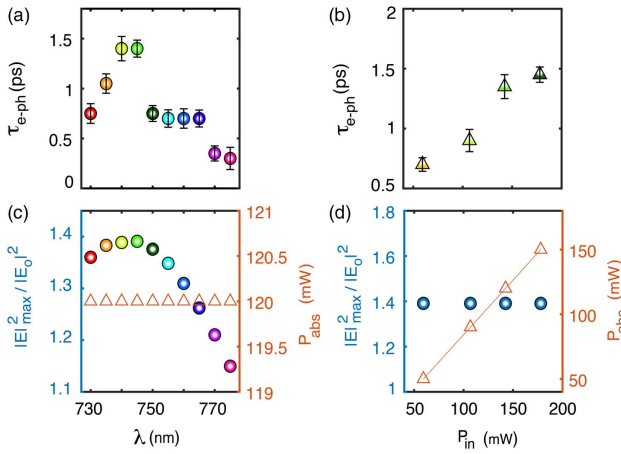
$$C_l \frac{\partial T_l}{\partial t} = G(T_e - T_l),$$



**Fig. 3.** (a) Differential reflectivity contour plot computed from the free electron model and transfer matrix methods. Hot-electron temperature as a function of the delay time between pump and probe beams under (b) fixed (120 mW) absorbed power and (c) fixed resonance wavelength (745 nm). The solid lines are the calculated electron temperatures, and the open circles are the electron temperatures obtained from our differential reflectivity measurements.

where  $T_e$  and  $T_l$  are electron and lattice temperature [45,46],  $C_e(T_e) = \frac{\pi^2 N_e k_b}{2} (k_b T_e / E_f)$  and  $C_l = 2.5 \times 10^6 \text{ J m}^{-3} \text{ K}^{-1}$  are the electron and lattice heat capacities [45,47],  $E_f$  and  $k_b$  are the Fermi level and Boltzmann constant, respectively,  $K_e = 315 \text{ W m}^{-1} \text{ K}^{-1}$  is the electron thermal conductivity, and  $G = C_e(T_e) / \tau_{e-ph}$  is the electron-phonon coupling coefficient within the weak perturbation approximation with  $\tau_{e-ph}$  as the electron-phonon relaxation time. Under the weak perturbation regime ( $T_e \ll T_f \sim 10^4$ ), where  $T_f$  is the Fermi temperature, the electrons' heat capacity is much smaller than the lattice heat capacity; this makes the lattice temperature relatively constant with respect to the electrons' temperature.

In general, the skin depth of a material is simply applied to the laser heating source term ( $S(z, t)$ ) to model the laser interaction with the material as a function of depth. Here, we modify the source term ( $S(z, t)$ ) by using the decaying length of the confined electric field of the surface plasmon on both sides of the interface instead of skin depth of the Au (see Fig. S2 of Supplement 1 for the electrical field profile of Au thin film). Using the finite difference time domain (FDTD) simulation, the electric field profile is numerically computed within the sample throughout the range of the wavelengths. To keep the total absorbed power constant, we vary the input power accordingly. The calculated field is fitted with double exponential terms, including the decaying field at the Au-prism interface and the decaying field at the Au-air interface. The modified source term ( $S(z, t)$ ) to incorporate the absorbed power profile inside the Au film can be described as



**Fig. 4.** Effect of field enhancement on relaxation time due to the surface plasmon coupling under fixed (120 mW) and variable (50 mW, 90 mW, 120 mW, 150 W) absorbed powers. Experimentally measured hot-electron relaxation time under (a) fixed and (b) variable absorbed powers. Field enhancement computed from the FDTD simulation for wavelengths ranging from 730 nm to 775 nm under (c) fixed and (d) variable absorbed powers. The electric field profiles are normalized by the intensity of the input field.

$$S(z, t) = \sqrt{\frac{\beta(1-R)\Phi}{\pi}} \frac{1}{t_p} \times \left( \frac{a_1}{b_1} e^{-z/b_1} + \frac{a_2}{b_2} e^{(z-d)/b_2} \right) e^{-\beta((t-2t_p)/t_p)^2}, \quad (2)$$

where  $t_p$  is the laser pulse width,  $\Phi$  is the laser fluence,  $d$  is the sample thickness, and  $\beta = 4\ln(2)$  [46].  $a_1$  and  $a_2$  correspond to the intensity of electric field at Au–air and Au–prism, and  $b_1$  and  $b_2$  correspond to the decaying length of electric field at Au–air and Au–prism, respectively.

Using our experimental conditions with the modified TTM, we numerically calculate the electron temperature as displayed in Figs. 3(b) and 3(c). For the case of constant absorbed power, we show four wavelengths and their corresponding best fits on the relaxation time to preserve space. The complete set of 10 wavelengths is presented in Fig. S3 of Supplement 1. We also incorporate the spatial dependence of the electron temperature by averaging the temperature profiles along the  $z$  direction (Fig. S4 of Supplement 1). The result of the fits is shown in Figs. 3(b) and 3(c) based on the normalized minimum squared error (NMSE) calculation for the hot-electron relaxation time. The good agreement of the calculated maximum temperature by TTM and the converted maximum temperature by the free electron model indicates that our free electron model is described well by electron temperature, because the calculated temperature using the TTM depends solely on the experimental conditions.

Figures 4(a) and 4(b) present the extracted hot-carrier relaxation time for cases of both fixed absorbed power and fixed illumination wavelength. When the incident power is varied while coupling to the surface plasmon [Fig. 4(b)], the hot-carrier relaxation time increases linearly with increasing incident pump power [Figs. 4(b) and 4(d)]. However, when the absorbed power is held constant and the internal field intensity profile is varied (i.e., the amount of surface plasmon coupling is varied), we find that the hot-carrier relaxation time is strongly dependent on the

intensity of the electric field [see the trend of hot-carrier relaxation time in Fig. 4(a) and the normalized maximum intensity of electric field in Fig. 4(c)]. This result confirms that the surface plasmon coupling can enhance the hot-carrier relaxation time in the Au film with high field confinement as well as the increase of the light absorption in the Au film. Notably, we can more effectively increase the hot-carrier relaxation time with the local electric field enhancement than with increasing the input power. We achieve approximately a doubling of the hot-carrier relaxation time with only a  $\sim 3.5\%$  increase in electric field intensity (normalized to the input field) at the metal–air interface through SP coupling. Although the hot-electron relaxation time and the corresponding e-ph coupling factor have been studied extensively as a function of the elevated electrons' temperature [48,49], the effect of electric field confinement on the relaxation time has not been fully determined. Furthermore, we hypothesize that electric field confinement could affect the reabsorption rate of the nonequilibrium phonon population due to a bottleneck effect. The increase in the reabsorption rate leads to the reduction of the thermalization rate and enhances the hot-electron's relaxation time, which has also been observed in the case of other high-density materials [50]. Consequently, this feature suggests that electric field confinement helps to excite free electrons to higher energy states, and these nonequilibrium hot electrons take longer to relax via a series of e-ph scattering processes.

### 3. CONCLUSION

In summary, we have experimentally demonstrated the impact of propagating surface plasmon excitation on the hot-carrier relaxation time through the use of a degenerate pump–probe technique under the Kretschmann configuration. We introduce an approach to analyze the unique internal field confinement in Au thin films with surface plasmon coupling by modifying the TTM. From the comparison study between the constant absorbed pump power and the constant electric field, we determine that electric field confinement results in the generation of long-lived hot electrons in the Au thin film. Our results provide a foundation for the design of efficient plasmonic systems to tailor hot-carrier lifetime with low power consumption in hot-carrier-based optoelectronic devices.

**Funding.** National Science Foundation (ECCS-1554503); Office of Naval Research (N00014-16-1-2540).

**Disclosures.** The authors declare no competing financial interests.

See Supplement 1 for supporting content.

†These authors contributed equally to this work.

### REFERENCES

1. M. L. Brongersma, N. J. Halas, and P. Nordlander, "Plasmon-induced hot carrier science and technology," *Nat. Nanotechnol.* **10**, 25–34 (2015).
2. A. J. Nozik, "Quantum dot solar cells," *Physica E* **14**, 115–120 (2002).
3. S. Mubeen, J. Lee, N. Singh, S. Krämer, G. D. Stucky, and M. Moskovits, "An autonomous photosynthetic device in which all charge carriers derive from surface plasmons," *Nat. Nanotechnol.* **8**, 247–251 (2013).

4. P. Reineck, D. Brick, P. Mulvaney, and U. Bach, "Plasmonic hot electron solar cells: the effect of nanoparticle size on quantum efficiency," *J. Phys. Chem. Lett.* **7**, 4137–4141 (2016).
5. A. J. Leenheer, P. Narang, N. S. Lewis, and H. A. Atwater, "Solar energy conversion via hot electron internal photoemission in metallic nanostructures: efficiency estimates," *J. Appl. Phys.* **115**, 1–7 (2014).
6. S. Peruch, A. Neira, G. A. Wurtz, B. Wells, V. A. Podolskiy, and A. V. Zayats, "Geometry defines ultrafast hot-carrier dynamics and Kerr non-linearity in plasmonic metamaterial waveguides and cavities," *Adv. Opt. Mater.* **5**, 1–12 (2017).
7. M. M. Jjadi, J. C. Konig-Otto, S. Winnerl, A. B. Sushkov, H. D. Drew, T. E. Murphy, and M. Mittendorff, "Nonlinear terahertz absorption of graphene plasmons," *Nano Lett.* **16**, 2734–2738 (2016).
8. M. Mehdi Jjadi, K. M. Daniels, R. L. Myers-Ward, D. K. Gaskill, J. C. Ko Nig-Otto, S. Winnerl, A. B. Sushkov, H. Dennis Drew, T. E. Murphy, and M. Mittendorff, "Optical control of plasmonic hot carriers in graphene," *ACS Photon.* **6**, 302–307 (2019).
9. L. J. Kraye, E. M. Tennyson, M. S. Leite, and J. N. Munday, "Near-IR imaging based on hot carrier generation in nanometer-scale optical coatings," *ACS Photon.* **5**, 306–311 (2018).
10. H. Chalabi, D. Schoen, and M. L. Brongersma, "Hot-electron photodetection with a plasmonic nanostripe antenna," *Nano Lett.* **14**, 1374–1380 (2014).
11. A. Sobhani, M. W. Knight, Y. Wang, B. Zheng, N. S. King, L. V. Brown, Z. Fang, P. Nordlander, and N. J. Halas, "Narrowband photodetection in the near-infrared with a plasmon-induced hot electron device," *Nat. Commun.* **4**, 1643 (2013).
12. P. N. Mark, W. Knight, H. Sobhani, and N. J. Halas, "Photodetection with active optical antennas," *Science* **332**, 702–704 (2011).
13. G. Baffou and R. Quidant, "Thermo-plasmonics: using metallic nanostructures as nano-sources of heat," *Laser Photon. Rev.* **7**, 171–187 (2013).
14. P. Zilio, M. Dipalo, F. Tantussi, G. C. Messina, and F. De Angelis, "Hot electrons in water: injection and ponderomotive acceleration by means of plasmonic nanoelectrodes," *Light Sci. Appl.* **6**, 1–8 (2017).
15. M. Kim, M. Lin, J. Son, H. Xu, and J. M. Nam, "Hot-electron-mediated photochemical reactions: principles, recent advances, and challenges," *Adv. Opt. Mater.* **5**, 1–21 (2017).
16. E. Cortés, W. Xie, J. Cambiasso, A. S. Jermyn, R. Sundararaman, P. Narang, S. Schlücker, and S. A. Maier, "Plasmonic hot electron transport drives nano-localized chemistry," *Nat. Commun.* **8**, 1–10 (2017).
17. M. S. Gerardo, A. Lopez, M.-C. Estevez, and L. M. Lechuga, "Recent advances in nanoplasmonic biosensors: applications and lab-on-a-chip integration," *Nanophotonics* **6**, 123–136 (2016).
18. J. N. Anker, W. P. Hall, O. Lyandres, N. C. Shah, J. Zhao, and R. P. V. Duyne, "Biosensing with plasmonic nanosensors," *Nat. Mater.* **7**, 442–453 (2008).
19. P. Christopher and M. Moskovits, "Hot charge carrier transmission from plasmonic nanostructures," *Annu. Rev. Phys. Chem.* **68**, 379–398 (2017).
20. G. Tagliabue, A. S. Jermyn, R. Sundararaman, A. J. Welch, J. S. DuChene, R. Pala, A. R. Davoyan, P. Narang, and H. A. Atwater, "Quantifying the role of surface plasmon excitation and hot carrier transport in plasmonic devices," *Nat. Commun.* **9**, 1–8 (2018).
21. P. Narang, R. Sundararaman, and H. A. Atwater, "Plasmonic hot carrier dynamics in solid-state and chemical systems for energy conversion," *Nanophotonics* **5**, 96–111 (2016).
22. A. Devižis, V. Vaičiškauskas, and V. Gulbinas, "Ultrafast pump-probe surface plasmon resonance spectroscopy of thin gold films," *Appl. Opt.* **45**, 2535–2539 (2006).
23. A. Devižis and V. Gulbinas, "Ultrafast dynamics of the real and imaginary permittivity parts of a photoexcited silver layer revealed by surface plasmon resonance," *Appl. Opt.* **47**, 1632–1637 (2008).
24. R. H. M. Groeneveld, R. Sprik, and A. Lagendijk, "Ultrafast relaxation of electrons probed by surface plasmons at a thin silver film," *Phys. Rev. Lett.* **64**, 784–787 (1990).
25. M. Z. Alam, I. De Leon, and R. W. Boyd, "Large optical nonlinearity of indium tin oxide in its epsilon-near-zero region," *Science* **352**, 795–797 (2016).
26. Y. Yang, K. Kelley, E. Sacht, S. Campione, T. S. Luk, J.-P. Maria, M. B. Sinclair, and I. Brener, "Femtosecond optical polarization switching using a cadmium oxide-based perfect absorber," *Nat. Photonics* **11**, 390–396 (2017).
27. G. V. Hartland, "Optical studies of dynamics in noble metal nanostructures," *Chem. Rev.* **111**, 3858–3887 (2011).
28. F. Wang and N. A. Melosh, "Plasmonic energy collection through hot carrier extraction," *Nano Lett.* **11**, 5426–5430 (2011).
29. M. Taghinejad, H. Taghinejad, Z. Xu, Y. Liu, S. P. Rodrigues, K. T. Lee, T. Lian, A. Adibi, and W. Cai, "Hot-electron-assisted femtosecond all-optical modulation in plasmonics," *Adv. Mater.* **30**, 1–7 (2018).
30. J. Kim, E. G. Carnemolla, C. Devault, A. M. Shaltout, D. Faccio, V. M. Shalaev, A. V. Kildishev, M. Ferrera, and A. Boltasseva, "Dynamic control of nanocavities with tunable metal oxides," *Nano Lett.* **18**, 740–746 (2018).
31. A. M. Brown, R. Sundararaman, P. Narang, W. A. Goddard, and H. A. Atwater, "Nonradiative plasmon decay and hot carrier dynamics: effects of phonons, surfaces, and geometry," *ACS Nano* **10**, 957–966 (2016).
32. M.-N. Su, C. J. Ciccarino, S. Kumar, P. D. Dongare, S. A. H. Jebeli, D. Renard, Y. Zhang, B. Ostovar, W.-S. Chang, P. Nordlander, N. J. Halas, R. Sundararaman, P. Narang, and S. Link, "Ultrafast electron dynamics in single aluminum nanostructures," *Nano Lett.* **19**, 3091–3097 (2019).
33. M.-N. Su, P. D. Dongare, D. Chakraborty, Y. Zhang, C. Yi, F. Wen, W.-S. Chang, P. Nordlander, J. E. Sader, N. J. Halas, and S. Link, "Optomechanics of single aluminum nanodisks," *Nano Lett.* **17**, 2575–2583 (2017).
34. F. He, N. Sheehan, S. R. Bank, and Y. Wang, "Giant electron-phonon coupling detected under surface plasmon resonance in Au film," *Opt. Lett.* **44**, 4590–4593 (2019).
35. R. H. M. Groeneveld, R. Sprik, and A. Lagendijk, "Femtosecond spectroscopy of electron-electron and electron-phonon energy relaxation in Ag and Au," *Phys. Rev. B* **51**, 11433–11445 (1995).
36. S. Link and M. A. El-Sayed, "Spectral properties and relaxation dynamics of surface plasmon electronic oscillations in gold and silver nanodots and nanorods," *J. Phys. Chem. B* **103**, 8410–8426 (1999).
37. S. Park, M. Pelton, M. Liu, P. Guyot-Sionnest, and N. F. Scherer, "Ultrafast resonant dynamics of surface plasmons in gold nanorods," *J. Phys. Chem. C* **111**, 116–123 (2007).
38. M. Ortolani, A. Mancini, A. Budweg, D. Garoli, D. Brida, and F. De Angelis, "Pump-probe spectroscopy study of ultrafast temperature dynamics in nanoporous gold," *Phys. Rev. B* **99**, 1–6 (2019).
39. S. I. Anisimov, B. L. Kapeliovich, T. L. Perel'man, and L. D. Landau, "Electron emission from metal surfaces exposed to ultrashort laser pulses," *J. Exp. Theor. Phys.* **39**, 375–377 (1974).
40. J. Hohlfeld, S.-S. Wellershoff, J. Güdde, U. Conrad, V. Jähnke, and E. Matthias, "Electron and lattice dynamics following optical excitation of metals," *Chem. Phys.* **251**, 237–258 (2000).
41. C.-K. Sun, F. Vallee, L. Acioli, E. P. Ippen, and J. G. Fujimoto, "Femtosecond investigation of electron thermalization in gold," *Phys. Rev. B* **48**, 12365–12368 (1993).
42. N. W. Ashcroft and N. D. Mermin, *Solid State Physics* (Holt, Rinehart and Winston, 1976).
43. N. Kinsey, C. DeVault, J. Kim, M. Ferrera, V. Shalaev, and A. Boltasseva, "Epsilon-near-zero Al-doped ZnO for ultrafast switching at telecom wavelengths," *Optica* **2**, 616–622 (2015).
44. B. T. Diroll, P. Guo, R. P. H. Chang, and R. D. Schaller, "Large transient optical modulation of epsilon-near-zero colloidal nanocrystals article," *ACS Nano* **10**, 10099–10105 (2016).
45. L. Jiang and H.-L. Tsai, "Improved two-temperature model and its application in ultrashort laser heating of metal films," *J. Heat Transfer* **127**, 1167–1173 (2005).
46. J. K. Chen and J. E. Beraun, "Numerical study of ultrashort laser pulse interactions with metal films," *Numer. Heat Transfer A* **40**, 1–20 (2001).
47. Z. Lin, L. V. Zhigilei, and V. Celli, "Electron-phonon coupling and electron heat capacity of metals under conditions of strong electron-phonon nonequilibrium," *Phys. Rev. B* **77**, 1–17 (2008).
48. N. Hogan, S. Wu, and M. Sheldon, "Photothermalization and hot electron dynamics in the steady state," *J. Phys. Chem. C* **124**, 4931–4945 (2020).
49. A. M. Brown, R. Sundararaman, P. Narang, W. A. G. Iii, and H. A. Atwater, "Ab initio phonon coupling and optical response of hot electrons in plasmonic metals," *Phys. Rev. B* **94**, 75120 (2016).
50. Y. Yang, D. P. Ostrowski, R. M. France, K. Zhu, J. Van De Lagemaat, J. M. Luther, and M. C. Beard, "Observation of a hot-phonon bottleneck in lead-iodide perovskites," *Nat. Photonics* **10**, 53–59 (2016).